

PCTWORLD INTELLECTUAL PROPERTY ORGANIZATION
International Bureau

INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(51) International Patent Classification ⁵ : G01N 27/12	A1	(11) International Publication Number: WO 93/20432 (43) International Publication Date: 14 October 1993 (14.10.93)
(21) International Application Number: PCT/GB93/00722 (22) International Filing Date: 6 April 1993 (06.04.93) (30) Priority data: 9207507.6 6 April 1992 (06.04.92) GB (71) Applicant (for all designated States except US): ELMWOOD SENSORS LIMITED [GB/GB]; Elm Road, North Shields, Tyne & Wear NE29 8SA (GB). (72) Inventor; and (75) Inventor/Applicant (for US only): HASHEMI, Tooraj [IR/GB]; 1 Patterdale Close, Belmont, Durham DH1 2UF (GB). (74) Agent: GARRATT, Peter, Douglas; Mathys & Squire, 10 Fleet Street, London EC4Y 1AY (GB).		(81) Designated States: CA, JP, US, European patent (AT, BE, CH, DE, DK, ES, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE). Published <i>With international search report.</i>
(54) Title: HUMIDITY SENSOR (57) Abstract A humidity sensor comprising a semi-conducting ceramic body having surface exposed to the atmosphere and whose electrical resistance varies with variation in the humidity of the atmosphere to which it is exposed, and means for measuring the resistance of the body and wherein said body comprises a semi-conducting ceramic composition comprising a solid solution of at least two metal oxides, at least one of which is a binary metal oxide having the formula $M_xM^1O_y$ where M is divalent metal, M^1 is tetravalent metal, x is 1 or 2 and y is $2 + x$ and a method of determining the level of humidity of an atmosphere using this humidity sensor.		

FOR THE PURPOSES OF INFORMATION ONLY

Codes used to identify States party to the PCT on the front pages of pamphlets publishing international applications under the PCT.

AT	Austria	FR	France	MR	Mauritania
AU	Australia	GA	Gabon	MW	Malawi
BB	Barbados	GB	United Kingdom	NL	Netherlands
BE	Belgium	GN	Guinea	NO	Norway
BF	Burkina Faso	GR	Greece	NZ	New Zealand
BG	Bulgaria	HU	Hungary	PL	Poland
BJ	Benin	IE	Ireland	PT	Portugal
BR	Brazil	IT	Italy	RO	Romania
CA	Canada	JP	Japan	RU	Russian Federation
CF	Central African Republic	KP	Democratic People's Republic of Korea	SD	Sudan
CC	Congo	KR	Republic of Korea	SE	Sweden
CH	Switzerland	KZ	Kazakhstan	SK	Slovak Republic
CI	Côte d'Ivoire	LJ	Liechtenstein	SN	Senegal
CM	Cameroon	LK	Sri Lanka	SU	Soviet Union
CS	Czechoslovakia	LU	Luxembourg	TD	Chad
CZ	Czech Republic	MC	Monaco	TG	Togo
DE	Germany	MG	Madagascar	UA	Ukraine
DK	Denmark	ML	Mali	US	United States of America
ES	Spain	MN	Mongolia	VN	Viet Nam
FI	Finland				

HUMIDITY SENSOR

This invention relates to improved sensors and to a method of determining the level of humidity of an atmosphere.

It is known that certain ceramic materials exhibit a change in resistance or impedance with changes in the humidity of the atmosphere to which they are exposed. Thus, these ceramic materials have been proposed for use in humidity sensors.

Humidity sensors comprising such ceramic materials offer advantages over conventional hygrometers of the kind which rely on changes of capacitance with change in humidity because changes in resistance are easier to monitor and register electronically.

IEEE Transactions on Electron Devices, Vol 1 ED-29, No. 1, January 1993 describes humidity sensors comprising $\text{MgCr}_2\text{O}_4 - \text{TiO}_2$ and $\text{Ba}_{2x}\text{Sr}_x\text{TiO}_3$ doped with MgCr_2O_4 . However, in use, the base resistance of the ceramic material gradually increases, necessitating a heat cleaning process between successive operations.

Another known humidity sensitive ceramic is Nb_2O_5 doped with TiO_2 . However, as described in Sensors and Actuators 24(1990) 55-60, the ceramic shows a gradual increase in impedance with time. Thus an aging process is required to produce a stable sensitivity to humidity.

Sensors and Actuators 4(1983) 85-96 describes how humidity sensitive ceramics $\text{MgCr}_2\text{O}_4 - \text{TiO}_2$, $\text{TiO}_2 - \text{V}_2\text{O}_5$ and Fe_2O_3 have been observed to show a progressive drift in resistance when exposed to highly humid environments.

The problems of drift in resistance and the need for heat cleaning have been overcome in some humidity sensors by the addition of small amounts of alkali ions. However these then produce problems with the stability of the alkali ions at the surface of the sensing materials. Such materials are MgFe_2O_4 doped with 2 mol% K^+ , Li^+ or Na^+ and $\text{ZnCr}_2\text{O}_4 - \text{LiZnVO}_4$ as described in Sensors and Actuators 10(1986) 379-398.

A MgAl_2O_4 sensor, as described in Sensors and Actuators 4 (1983) 85-96, has good humidity sensitivity. However, the impedance in less humid atmospheres was found to be too high for practical measurements.

A sensor consisting of SnO_2 with 10% Al_2O_3 and 5% TiO_2 was found to have shortcomings due to significant temperature dependence and sensitivity to inflammable gases as described in Sensors and Actuators 16(1989) 1-11.

The materials CaMnO_3 , SrFeO_3 , BaTiO_3 , LaCrO_3 and $\text{Ba}_{0.998}\text{Sb}_{0.002}\text{TiO}_3$ were found to be insensitive to humidity changes. CaTiO_3 , CaSnO_3 , SrTiO_3 and SrSnO_3 showed humidity sensitivity at elevated temperature but had conductivities too small for practical use as humidity sensors. The materials $\text{Co}_{0.9}\text{La}_{0.1}\text{TiO}_3$, $\text{Ca}_{0.9}\text{La}_{0.1}\text{SnO}_4$, $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_3$ and $\text{Sr}_{0.9}\text{La}_{0.1}\text{SnO}_3$ showed a performance which was strongly dependent on temperature and a response time which was slow - taking 5 minutes to attain steady-state values of conductivity:

A K_2O doped $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{TiO}_3$ humidity sensor showed an inductance loop over low relative humidity (5-30%) and low frequency (5-40 Hz) range making it unsuitable for use under those conditions.

As described in Journal of Material Science 26(1991) 3874-3878, $\text{Ni}(\text{Al}, \text{Fe})_2\text{O}_4 - \text{TiO}_2$ sensors showed slower response to decreasing humidity than to increasing humidity above 40%. Thus a hysteresis loop is present.

We have discovered that a semi-conducting ceramic composition comprising a solid solution of at least two metal oxides, at least one of which being a binary metal oxide having the formula $\text{M}_x\text{M}'\text{O}_y$ where M is divalent metal, M' is tetravalent metal, x is 1 or 2 and y is $2 + x$, is superior to known ceramics for use in humidity sensors since they do not exhibit the disadvantages and drawbacks of the known ceramics described above.

In addition, the compositions of the present invention generally exhibit chemical stability within the range of temperatures at which atmospheric humidity is normally required to be measured and furthermore, the porosity of molded forms of the

compositions, which is important in a humidity sensor since it controls the amount of surface area available to the atmosphere for a given volume or mass, is readily controllable by use of appropriate means for forming the molded body, e.g. powder pressing and/or sintering.

For the purposes of this specification, Si and Ge are classed as metals. In addition, for the purposes of the specification, where we refer to a solid solution of at least two metal oxides we mean either doping at least one metal oxide with a small amount of at least one other metal oxide, or a true solid solution. The ability of two substances to form solid solutions is determined primarily by geometrical considerations but also by electronic and other factors.

There are two primary ways by which "foreign" species may be introduced into crystal structures as a solid solution with the simultaneous maintenance of thermodynamic equilibrium: (a) the direct substitution of the solute for the host atom, (b) the introduction of the solute into normally unfilled interstitial positions.

In the direct substitutional solid solution (a), the solute cations or anions (or both) substitute in a random manner directly onto the crystal structure of solvent ions; the extent of solution is primarily governed by certain structural and chemical factors.

In interstitial solid solutions (b), the solute ions are normally small and can occupy, normally in limited amounts, interstitial positions rather than substitute for the filled sites of the solvent material.

According to the present invention there is provided a humidity sensor comprising a semi-conducting ceramic body having surface exposed to the atmosphere and whose electrical resistance varies with the variation in the humidity of the atmosphere to which it is exposed, and means for measuring the resistance of the body and wherein said body comprises a semi-conducting ceramic composition comprising a solid solution of at least two metal oxides, at least one of which is a binary metal oxide having the formula $M_xM^yO_y$, where M is divalent metal, M^y is tetravalent metal, x is 1 or 2 and y is $2 + x$.

The invention also provides a method of determining the level of humidity of an atmosphere, the method comprising measuring the resistance of a semi-conducting ceramic body comprising a solid solution of at least two metal oxides, at least one of which being a binary metal oxide having the formula $M_xM^yO_y$, where M is divalent metal, M^y is tetravalent x is 1 or 2 and y is $2 + x$, the surface of said solid solution being exposed to said atmosphere.

In general, it will be found that in order to obtain a solid solution the metals of the simple metal oxides and of the binary metal oxides should be chosen from Groups IB, IIB, IVB and VIII of the Periodic Table, and in particular from those which also fall in the 4th and 5th Periods of the Table. All references herein to the Periodic Table relate to the version found on page 149 of "Table of Physical and Chemical Constants" by G.W.C. Kaye and T.H. Laby, published by Longman, 14th Edition.

Specific examples of simple metal oxides suitable for use in this invention are ZnO, CoO, TiO, Co_3O_4 and NiO. Specific examples of binary metal oxides suitable for use in the invention are Zn_2SiO_4 , Co_2SiO_4 , Ni_2SiO_4 , Zn_2GeO_4 , $CuGeO_3$, Co_2GeO_4 , Ni_2GeO_4 , Zn_2SnO_4 , $MgSnO_4$, Co_2SnO_4 , Cd_2SnO_4 , Ba_2SnO_4 , Ni_2SnO_4 .

Specific examples of combinations of oxides which form solid solutions suitable for use in the invention are

Combination		Ranges of concentrations at which solid solution is obtainable (mole %)
A	B	
Zn_2GeO_4	Co_2GeO_4	up to 19% B in A or up to 35% A in B
Zn_2GeO_4	Ni_2GeO_4	less than 19% B in A or up to 14% A in B
Zn_2SiO_4	Co_2SiO_4	up to 50% B in A or up to at least 10% A in B
Co_2GeO_4	Ni_2GeO_4	entire range of concentration
Zn_2SnO_4	Co_2SnO_4	expected to be entire range of concentration
Zn_2GeO_4	$CuGeO_3$	
Zn_2GeO_4	Zn_2SiO_4	entire range of concentration
Zn_2SnO_4	NiO	

Zn_2SnO_4	CoO
Co_2GeO_4	Co_2SiO_4
Co_2GeO_4	Co_2SnO_4
Co_2SiO_4	Co_2SnO_4
Zn_2GeO_4	Zn_2SnO_4
Zn_2SiO_4	Zn_2SnO_4
Ni_2GeO_4	Ni_2SiO_4

It is not necessary for the entire composition to be in the form of a solid solution, provided that at least a part of the mixture of oxides has formed a solid solution. The present of a solid solution in the composition may be detected by X-Ray analysis in known manner.

The composition comprising a solid solution may be formed in any suitable manner. One preferred method involves mixing the oxides and then firing or sintering the mixture. The composition may be formed into a ceramic body such as a pellet or wafer either during or after the step of forming the solid solution. Where the electrical resistance of the as-formed body is too high for the intended use, this may be lowered by annealing in a reducing atmosphere. This means that by appropriate choice and control of heat treatment, electrical resistivity may be adjusted to meet any particular need.

Without wishing to be bound by the following theory, we believe that when the ceramic is of an ionic-type the change in resistance with humidity is due to chemisorption of water molecules on the surface of the oxides. The chemisorption causes dissociation of water molecules to form surface hydroxyls. Physisorption can then take place on top of this.

When the ceramic is of an electronic - type, the water molecules in the atmosphere being detected act as an electron donating gas on the semi-conductive oxides and thereby increase conductivity.

With increased humidity, further layers of water are adsorbed and the high electrostatic fields resulting from the initially adsorbed species causes a far greater dissociation and

therefore to a further reduction in resistivity.

At drier conditions, the number of extra charge carriers fall to the extent that the resistivity is dominated by the ceramic bulk.

The invention will now be illustrated by the following Examples and drawings in which:

Figure 1 is an X-Ray diffraction pattern of the ceramic formed in Example 1.

Figure 2 is a plot of resistance against atmospheric humidity of the ceramic formed in Example 2.

Figure 3 is a plot of resistance against atmospheric humidity of the ceramic formed in Example 3.

Figure 4 is a plot of $\log(\text{resistance})$ against relative humidity for the ceramic formed in Example 14.

Figure 5 is a plot of $\log(\text{resistance})$ against relative humidity for the ceramic formed in Example 16.

Figure 6 is a schematic diagram of one arrangement for the sensor.

Example 1

In a first experiment, an intimate mixture of ZnO and GeO_2 , in the molar ratio of two of ZnO to one of GeO_2 , was formed by one of two alternative methods.

In the first a slurry was formed from the powder mixture and de-ionized water. The slurry which was then ball-milled with cylindrical sialon media for approximately 12 hours. The resulting slurry was then dried in an oven to evaporate the water and the

residue was ground to powder form in a pestal and mortar. In the second method the powder was mixed by hand in a pestal and mortar. The first of these two methods produces a more intimate mixture of the two starting materials and a finer particle size due to the crushing effect of the ball-milling. Ball-milling, however, is time consuming and best suited to large quantities of powder mix (100-200g). In most experimental cases smaller amounts of each sample were produced (20-30g). Since it was found that the intimacy of mixing and the particle size produced by pestal and mortar mixing was sufficient to produce monophasic, well sintered material from the firing phase, this method was used in preference in most cases.

The mixture was then pressed isostatically into 13mm diameter 1mm thick discs and then fired at 1200°C for 12 hours, thus the calcination and sintering stages were carried out in a single firing process. The sintered bodies exhibited up to 4% expansion in volume. The pellets were then crushed, reground with about 10% cobaltic oxide (Co_3O_4), pressed again into 13mm diameter 1mm thick discs and fired for another 12 hours at 1200°C. The Co_3O_4 is converted to CoO during firing.

In a second experiment an intimate mixture of the three oxides in the same proportions was pressed into 13mm diameter 1mm thick discs and fired for 12 hours at 1200°C.

The X-Ray diffraction patterns of the samples showed Zn_2GeO_4 and ZnO indicating that a reaction between the cobalt oxide and zinc germanate has occurred resulting in the formation of cobalt germanate and zinc oxide and that formation of a solid solution between the cobalt germanate and Zn_2GeO_4 has occurred. The X-Ray diffraction patterns are reproduced in Figure 1.

Semi-conductivity was then induced in the product by annealing in a dry nitrogen atmosphere. Resistance versus temperature characteristics of the samples were studied using a purpose built microprocessor controlled system in which two point resistance measurements could be performed on up to 10 samples with temperature controlled from room temperature to 450°C.

Gallium-indium contacts were applied to the top and bottom surfaces of the sintered pellets, which were first smoothed with wet and dry paper. A DC potential of 2 volts

was applied across the sample and the resulting current was measured, whilst the sample was heated in air in a computer controlled oven to 450°C at a rate of 3°C per minute. Current measurements were made at temperature increments of 10°C with the sample first held at the required temperature for 2 minutes to allow the current to stabilize. The measurements of temperature and current were recorded by the computer.

The resistance versus relative humidity characteristics were investigated using a purpose built humidity chamber, an electrometer and a commercial humidity meter.

The humidity chamber consisted of a sealed cylindrical chamber with two inlet valves, a gas mixing chamber between them, an outlet valve, a recess into which was sealed the probe from the humidity meter, a copper stage with a thermocouple and temperature control circuit and electrical connections for two point resistance measurements on four samples. The two inlet valves were connected to dry and saturated gas supplies respectively. Bottled nitrogen was used for the dry gas and compressed air bubbled through water at room temperature for the saturated gas. The humidity in the chamber was controlled by allowing a mixture of the two gases into the chamber, whilst allowing air out via the outlet valve. The humidity in the chamber was measured using a Thermal Measurement Systems THP 338 digital Thermo-Hygro recorder. A Keithley 617 programmable electrometer was used to provide a constant voltage across the two contacts on the sample and to measure the resulting current. Resistance of the samples was in the range of 100's MΩ so current values to be measured were very small. Gallium-indium contacts were used.

Voltage current characteristics were measured from 0–100V. Current was measured with respect to time for constant of voltage and humidity to check the stability of the samples. Current versus relative humidity was measured, keeping values of voltage and temperature constant.

Measurement of the resistance of the body while exposing it to atmosphere of varying humidity at a given temperature proved that the resistance was highly dependent on the relative humidity. For example, a change of resistivity was found to occur from 5×10^7 ohm in ambient atmosphere to about 10^{12} ohm in vacuum or dry atmosphere.

This property was found to be reproducible and reversible.

Example 2

Following the pressing and sintering procedure described in Example 1, a semi-conductive ceramic solid solution of 16 mole % Zn_2GeO_4 and 84 mole % Ni_2GeO_4 was formed and a plot of its resistance against atmospheric humidity, expressed as the difference in temperature between wet bulb and dry bulb, is reproduced as Figure 2.

Example 3

A semi-conductive ceramic solid solution of 90.5 mole % Zn_2GeO_4 and 9.5 mole % Co_2GeO_4 was formed following the pressing and sintering procedure described in Example 1 and using a sintering temperature of 1200°C , and a plot of its resistance against atmospheric humidity is reproduced as Figure 3.

Example 4

A semi-conductive doped solid solution was formed in accordance with the procedure described in Example 1 except that the sintered pellets were crushed and ground to a fine powder. The material and required amount of dopant were then mixed thoroughly in a pestal and mortar, as with the precursor powders. The doped material was then pressed into pellets using a small quantity of PVA binder, since once fired the material proved to be too dry to adhere in the green-body form. These pellets were then re-fired in air for 12 hours at 1200°C .

Examples 5 to 39

Following the pressing and sintering procedure described in Example 1 and using a sintering temperature of 1200°C , semi-conductive ceramic solid solutions were formed from the following combinations of oxides and found to exhibit electrical resistivity which varied with variation in the humidity of the atmosphere to which the ceramic body was exposed. In all the following examples, the reported percentages are molar unless otherwise indicated.

Example	Composition
5	90.5% Zn_2GeO_4 and 9.5% Co_2GeO_4
6	91.3% Co_2GeO_4 and 8.7% Zn_2GeO_4
7	84% Co_2GeO_4 and 16% Zn_2GeO_4
8	91.3% Ni_2GeO_4 and 8.7% Zn_2GeO_4
9	84% Ni_2GeO_4 and 16% Zn_2GeO_4
10	90% Co_2GeO_4 and 10% Ni_2GeO_4
11	50% Co_2GeO_4 and 50% Ni_2GeO_4
12	10% Co_2GeO_4 and 90% Ni_2GeO_4
13*	90% Zn_2GeO_4 and 10% CuGeO_3
14	Zn_2GeO_4 + 10% Co_3O_4
15	Zn_2GeO_4 + 10% NiO
16	Zn_2GeO_4 + 10% Co_2GeO_4
17	Co_2GeO_4 + 10% Zn_2GeO_4
18	Zn_2GeO_4 + 10% Ni_2GeO_4
19	Ni_2GeO_4 + 10% Zn_2GeO_4
20	Zn_2SnO_4 + 10% Co_3O_4
21	Zn_2SnO_4 + 10% Co_2SnO_4
22	Zn_2SnO_4 + 10% NiO
23	Zn_2SiO_4 + 10% Co_3O_4
24	Zn_2SiO_4 + 10% Co_2SiO_4
25	Zn_2SiO_4 + 10% NiO
26	Zn_2SiO_4 + 10% Ni_2SiO_4
27	Zn_2SnO_4 + 10% Cd_2SnO_4
28	Zn_2SnO_4 + 10% MgSnO_3
29	Zn_2SnO_4 + 10% Ba_2SnO_4
30	Zn_2SnO_4 + 10% BaTiO_3
31	Zn_2SnO_4 + 10% TiO
32	MgSnO_4 + 10% Zn_2SnO_4
33	Cd_2SnO_4 + 10% Co_2SnO_4
34	Zn_2GeO_4 + 10% Zn_2SiO_4
35	Zn_2SnO_4 + 10% Co_2GeO_4

- | | |
|----|---|
| 36 | $\text{Zn}_2\text{GeO}_4 + 10\% \text{Co}_2\text{SiO}_4$ |
| 37 | Zn_2SnO_4 and 10% by weight, based on
Zn_2SnO_4 , of Co_2SnO_4 |
| 38 | Zn_2SiO_4 and 10% by weight, based on
Zn_2SiO_4 , of Co_2SiO_4 |
| 39 | Co_2SiO_4 and 10% by weight, based on
Co_2SiO_4 , of Zn_2SiO_4 |

* fired at 1000°C

Plots of log (resistance) against relative humidity for the compositions of Examples 14 and 16 are reproduced as Figures 4 and 5, respectively.

Plots of resistance against relative humidity show a decrease in resistance with increasing relative humidity. The decrease in resistance with humidity is approximately exponential, with the resistance levelling off at high and low humidity.

With reference to Figure 6, the sensor comprises electrodes 1, which are in contact with a pellet of the semiconducting ceramic body, 2. The electrolyte, 3, is formed by hydroxyl ions chemisorbed and/or physisorbed on the surface 4 of the ceramic body. The hydroxyl ions are formed by the disassociation of water molecules in the humid atmosphere.

CLAIMS

1. A humidity sensor comprising a semi-conducting ceramic body having surface exposed to the atmosphere and whose electrical resistance varies with variation in the humidity of the atmosphere to which it is exposed, and means for measuring the resistance of the body and wherein said body comprises a semi-conducting ceramic composition comprising a solid solution of at least two metal oxides, at least one of which is a binary metal oxide having the formula $M_xM^1O_y$ where M is divalent metal, M^1 is tetravalent metal, x is 1 or 2 and y is $2 + x$.
2. A humidity sensor as claimed in Claim 1, characterised in that the metals are selected from Groups IB, IIB, IVB and VIII of the Periodic Table
3. The humidity sensor as claimed in Claim 1 or 2 characterised in that the metals are selected from the 4th and 5th Period of the Periodic Table.
4. A humidity sensor as claimed in any one of Claims 1 to 3, characterised in that the ceramic composition comprises a solid solution of two metal oxides, both of which are binary metal oxides having the formula $M_xM^1O_y$ where M is divalent metal, M^1 is tetravalent metal, x is 1 or 2 and y is $2 + x$.
5. A humidity sensor as claimed in any one of Claims 1 to 4 characterised in that the ceramic composition is porous.
6. A humidity sensor as claimed in any one of Claims 1 to 5, characterised in that the ceramic body may be formed by powder pressing, sintering or powder pressing and sintering.
7. A method of determining the level of humidity of an atmosphere, the method comprising measuring the resistance of a semi-conducting ceramic body comprising a solid solution of at least two metal oxides, at least one of which is a binary metal oxide having the formula $M_xM^1O_y$ where M is divalent metal, M^1 is tetravalent metal, x is 1 or 2 and y is $2 + x$, a surface of said body being exposed to said atmosphere.

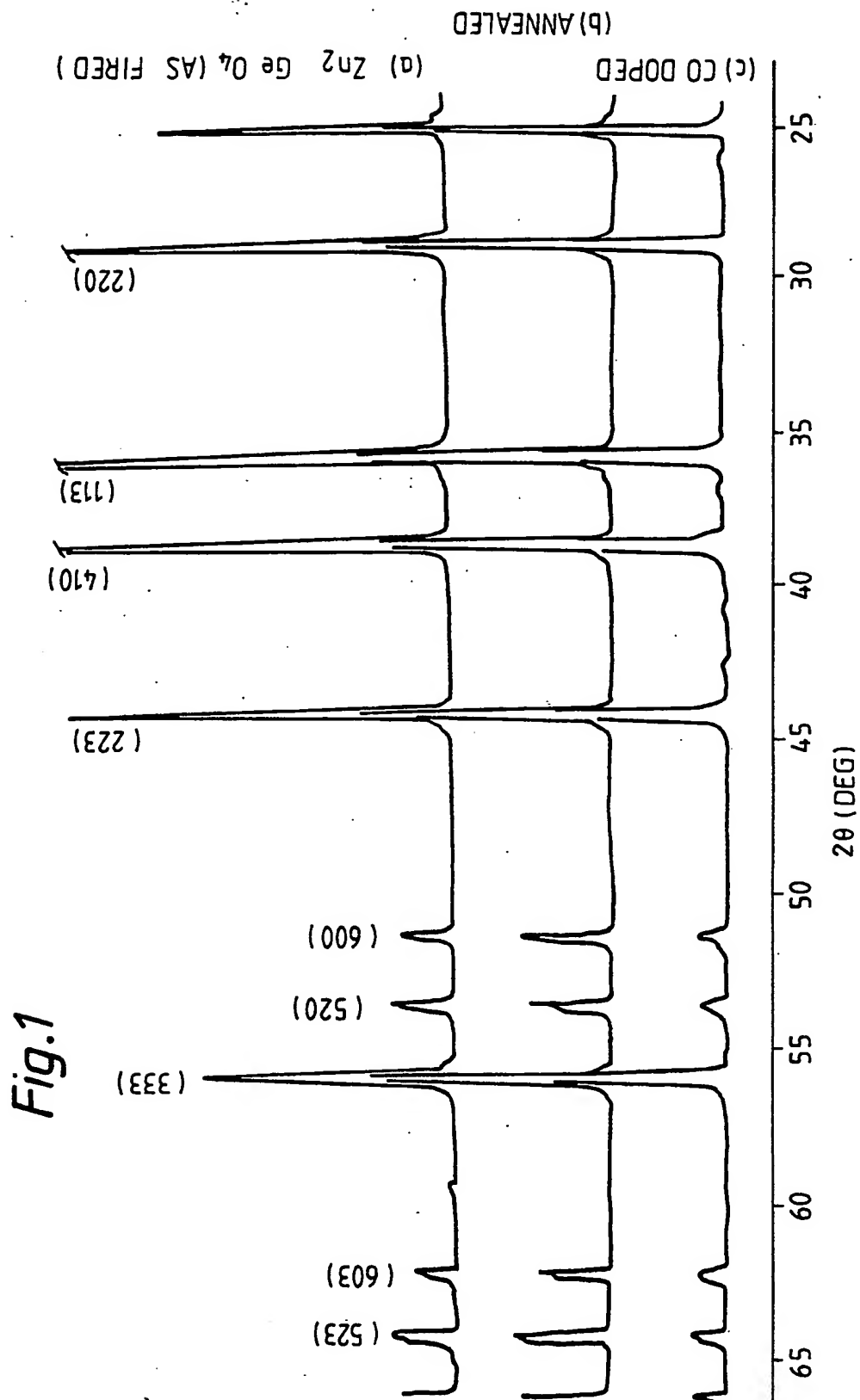


Fig.1

Fig. 2

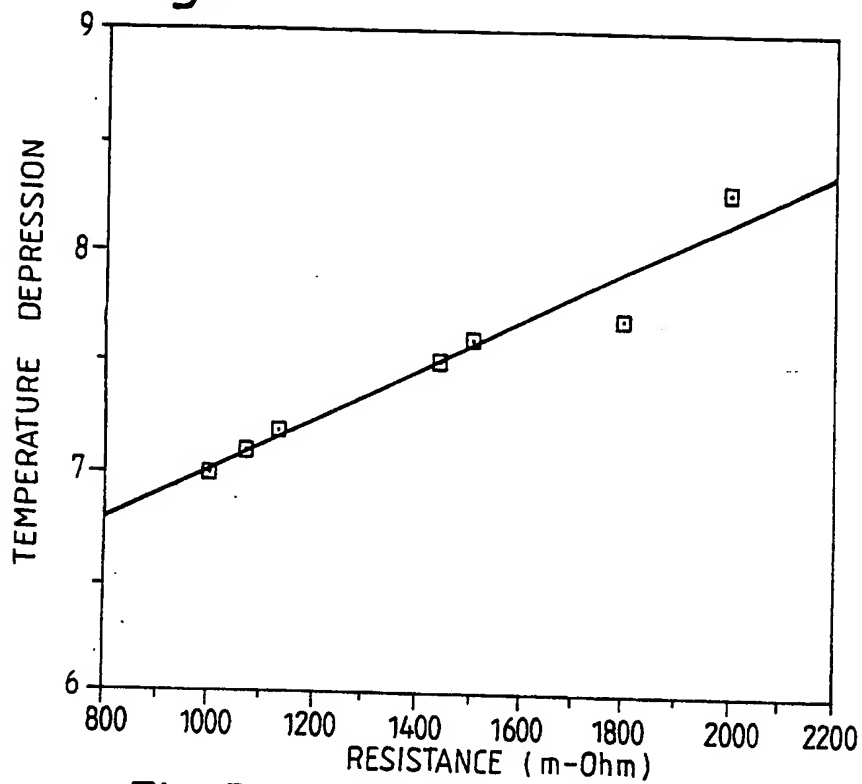


Fig. 3

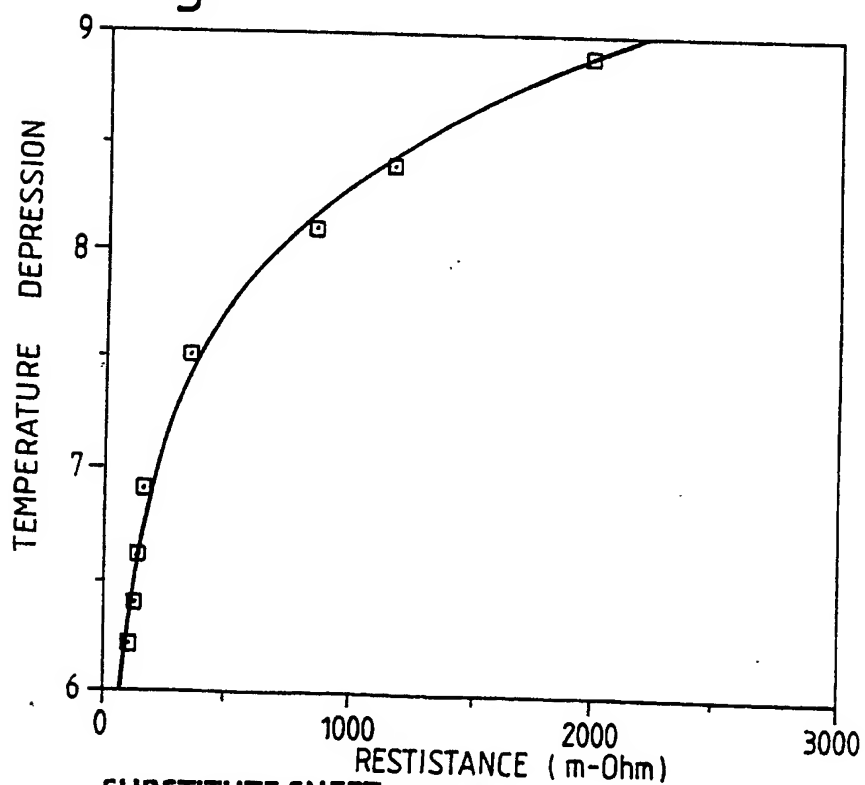


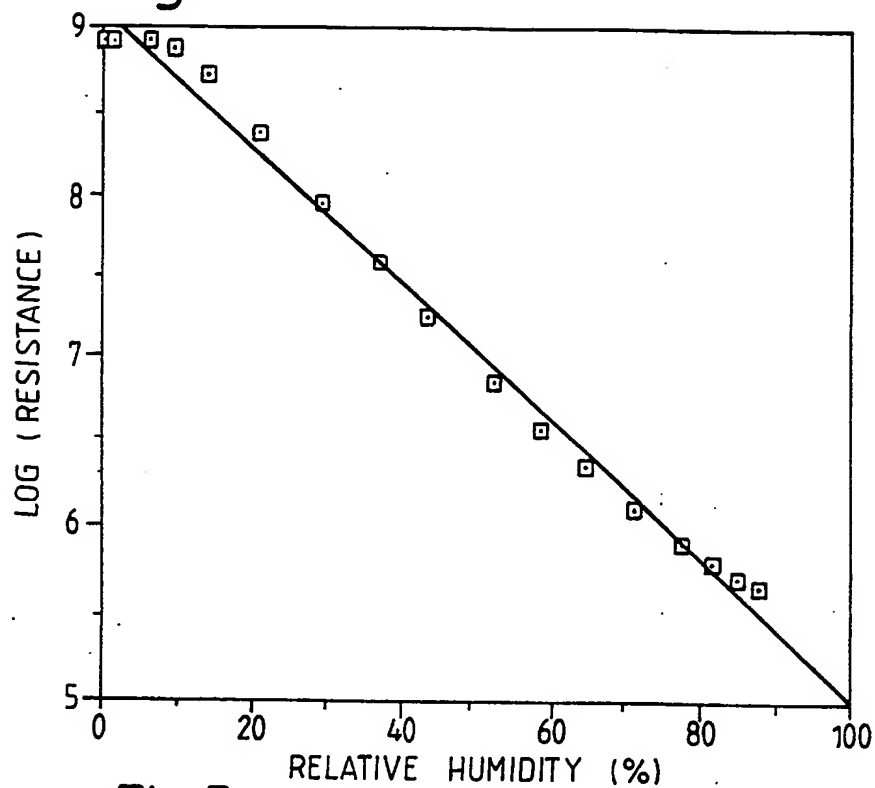
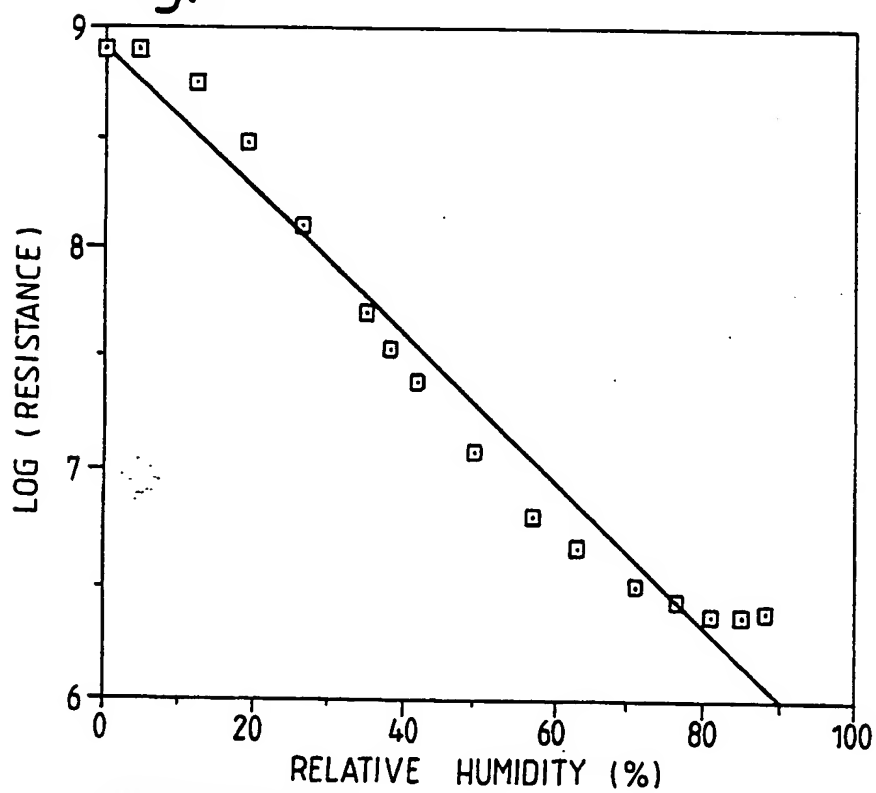
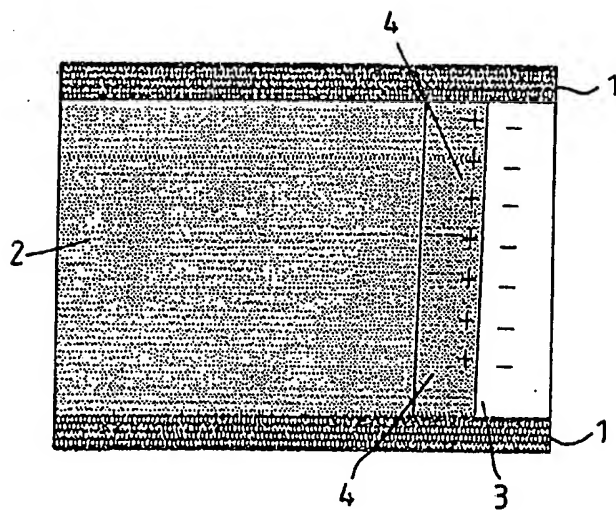
Fig.4*Fig.5*

Fig. 6



INTERNATIONAL SEARCH REPORT

PCT/GB 93/00722

International Application No

I. CLASSIFICATION OF SUBJECT MATTER (if several classification symbols apply, indicate all) ⁶		
According to International Patent Classification (IPC) or to both National Classification and IPC:		
Int.Cl. 5 G01N27/12		
II. FIELDS SEARCHED		
Minimum Documentation Searched ⁷		
Classification System	Classification Symbols	
Int.Cl. 5	G01N	
Documentation Searched other than Minimum Documentation to the Extent that such Documents are Included in the Fields Searched ⁸		
III. DOCUMENTS CONSIDERED TO BE RELEVANT⁹		
Category ¹⁰	Citation of Document, ¹¹ with indication, where appropriate, of the relevant passages ¹²	Relevant to Claim No. ¹³
X	DE,A,3 150 558 (MURATA MANUFACTURING C.) 19 August 1982 see the whole document ---	1-7
X	US,A,4 497 701 (MURATA ET AL.) 5 February 1985 see column 2, line 4 - column 2, line 16 ---	1-7
A	US,A,4 677 415 (HOWNG) 30 June 1987 see claim 1 ---	1-7
A	DE,A,3 821 620 (AISIN SEIKI K.K.) 5 January 1989 see the whole document -----	1-7
<p>¹⁰ Special categories of cited documents:</p> <p>"A" document defining the general state of the art which is not considered to be of particular relevance</p> <p>"E" earlier document but published on or after the international filing date</p> <p>"I" document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)</p> <p>"O" document referring to an oral disclosure, use, exhibition or other means</p> <p>"P" document published prior to the international filing date but later than the priority date claimed</p> <p>"T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention</p> <p>"X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step</p> <p>"Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art.</p> <p>"&" document member of the same patent family</p>		
IV. CERTIFICATION		
Date of the Actual Completion of the International Search		Date of Mailing of this International Search Report
23 JUNE 1993		07.07.93
International Searching Authority		Signature of Authorized Officer
EUROPEAN PATENT OFFICE		KOUZELIS D.

**ANNEX TO THE INTERNATIONAL SEARCH REPORT
ON INTERNATIONAL PATENT APPLICATION NO.**

GB 9300722
SA 72623

This annex lists the patent family members relating to the patent documents cited in the above-mentioned international search report.
The members are as contained in the European Patent Office EDP file on
The European Patent Office is in no way liable for these particulars which are merely given for the purpose of information. 23/06/93

Patent document cited in search report	Publication date	Patent family member(s)	Publication date
DE-A-3150558	19-08-82	JP-C- 1331385	14-08-86
		JP-A- 57106568	02-07-82
		JP-B- 60054259	29-11-85
		US-A- 4357426	02-11-82

US-A-4497701	05-02-85	None	

US-A-4677415	30-06-87	JP-A- 61260605	18-11-86

DE-A-3821620	05-01-89	JP-A- 1103803	20-04-89
		US-A- 4938892	03-07-90
